# Point defects in $Ga_2O_3$ as efficient UV V is light emission centers

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## Ga<sub>2</sub>O<sub>3</sub> ultra-wide-band-gap oxide

- $E_{a}$  = 4.8 eV  $\rightarrow$  a member of UWBGO family
- Transparent material; candidate for advanced technology applications (solar-blind UV photodetectors, solar cells, sensors and power electronic applications)
- 5 polymorphs (rhombohedral  $\alpha$ , monoclinic  $\beta$ , defected spinel  $\gamma$ , cubic  $\delta$  and orthorhombic
- Advantages over GaN and SiC: the cost of producing large-area substrates potentially lower (melt growth techniques possible), MOSFET with larger electric field
- Absence of clear demonstration of *p*-type doping (full  $Ga_2O_3$ -based bipolar devices are impossible)

## **Rapid Thermal Annealing @ 800°C**

- A significant difference between as grown and annealed  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystals
- Annealing at 800°C results in evolution of PL spectra  $\rightarrow$  PL fine structure, in which zero-phonon-line (ZPL) at 2.99 eV is accompanied by phonon repetitions separated by 145 meV
- High difference in Huang-Rhys factor





Gallium Oxide Wafer (Stanford Advanced Materials) //www.samaterials.com



 $\beta$ -Ga<sub>2</sub>O<sub>3</sub> monoclinic crystal structure

- Monoclinic crystal structure; 3 oxygen sites ( $O_1$ ,  $O_2$  two trigonal, and one  $O_2$  tetragonal) and 2 Ga sites (tetrahedral & octahedral)  $\rightarrow$  anisotropy of physical, optical and transport properties
- DFT calculations  $\rightarrow$  CBM at the  $\Gamma$  point and fairly flat VB;

indirect  $E_a = 4.66 \text{ eV}$ , direct  $E_a = 4.69 \text{ eV}$ 

- Large  $m_h$ ; holes tend to form small polarons localized at lattice imperfections
- Lack of near-band-edge (NBE) photoluminescence
- RT PL dominated by a broad set of transitions @ ~399nm ascribed to oxygen vacancies  $(V_{OI}, V_{OII} \text{ and } V_{OIII}) \rightarrow 416, 442 \text{ and } 464 \text{ nm}; \text{ but several mechanisms invoked to explain the}$ UVA PL (native defects, self-trapped polarons)
- Ultra-wide  $E_q \rightarrow$  attractive matrix for Rare Earth (RE) dopants

## Room temperature PL of $\beta$ -Ga<sub>2</sub>O<sub>3</sub>:RE

- $E_q$  of 4.9 eV  $\rightarrow$  transparent material, appropriate matrix for Rare Earth-based emitters
- Single β-Ga<sub>2</sub>O<sub>3</sub> (010) crystals (from Kyma Co.) implanted with Sm and Eu to a fluence of 4e14, 1e15 and 5e15 at/cm<sup>2</sup>, implantation energy 150 keV



### PL temperature dependence

- RTP annealed  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystals  $\rightarrow$  PL temperature dependence with E<sub>exc</sub>= 3.49 eV
- PL fine structure, with the ZPL at 2.99 eV accompanied by 3 phonon repetitions is observed at higher temperature



PL dynamics of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>  $\hbar \omega_{ev}$  =3.493 eV,  $\hbar \omega_{det}$  = 2.88 eV

#### $\beta$ -Ga<sub>2</sub>O<sub>3</sub> RTP 800°C. Temperature dependence @ $\hbar \omega_{ex}$ =3.49 eV



- PL temperature dependence with E<sub>exc</sub>= 3.49 eV (resonant excitation of PL line)
- Fast atypical PL drop for  $T \le 50K$  for RTP annealed  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystal and for T  $\leq$ 250K for as grown
- As grown  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystal  $\rightarrow$  interplay between different transitions; a few PL centers below 30K

#### PL decay time

- Implantation performed at UMCS (Marie Curie-Skłodowska University, Lublin, Poland, dr M. Turek)
- Intensive RE-related photoluminescence observed at RT
- WBG of 4.9 eV  $\rightarrow$  weak PL quenching expected; PL does not depend on temperature
- Samarium implantation with a fluence of  $4.10^{14}$ ,  $1.10^{15}$  and  $5.10^{15}$  at/cm<sup>2</sup>  $\rightarrow$  intensive PL at RT
- Intensive PL @ RT for high fluences  $\rightarrow$  quenching effect not observed



**Broad PL band @ RT** PL of RE doped  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>T=300K  $\hbar\omega_L$ =5.821 eV m - doped Eu - doped 10000 1.5 2.0 2,5 3,0 3.5 4.0 Energy (eV)

- Bright PL from defect states appears at 2.4 3.2 eV (with excitation above  $E_{\alpha}$ )
- Energy far from  $E_a = 4.9 \text{ eV} \rightarrow \text{intensive RT PL}$ (relatively weak dependence on the Fermi function)



- PL decay time measured for  $hv_{exc}$ = 3.49 eV and  $hv_{det}$ = 2.88 eV
- PL decay is non-exponential  $\rightarrow$  indirect indication of DAP centrum
- Relatively short decay times (~4 ns at RT) indicate isolated centrum  $\rightarrow$  good candidate for a single photon emitter (SPE)?

## **Density Functional Theory calculations**

- DFT: Calculation performed using QUANTUM-ESPRESSO package.
- Within the generalized gradient approximation (GGA)+U.
- Hubbard-like +U terms where applied on the both d(Ga) and p(O).
- $U_{Ga} = 3 \text{ eV}$ ,  $U_0 = 7 \text{ eV}$  give the correct band structure of  $Ga_2O_3$ . Calculated  $E_q = 4.85$  eV.
- Defect calculations in the 160-atoms supercell





- Total density of states of Ga<sub>2</sub>O<sub>3</sub>: without defects (pure) and involving one vacancy at  $Ga_1, Ga_2, O_1, O_2$  and  $O_3$  sites, respectively; Ga-vacancies are deep acceptor states; O-vacancies are deep donor states
- with different electronic structure that depends on site



• This PL band is very common and appears in undoped and implanted crystals



- Laser excitation above  $E_{\alpha}$  (5.82 eV)  $\rightarrow$  bright RT PL from defect states appears at 2.2 – 3.4 eV, with max at 2.95 eV (intensity weakly dependent on the Fermi function)
- PL excitation (PLE) with a xenon lamp shows max intensity of the 2.95 eV line at 5 eV (close to bandgap)
- Urbach states near the CB edge

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- Single  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> crystal is a good candidate for optical applications when doped with RE ions
- Single  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> does not show near-band-edge, but bright RT PL from defect states at 2.2 3.4 eV, with max at 2.95 eV
- Annealing at 800°C results in evolution of PL spectra  $\rightarrow$  PL fine structure, in which zero-phonon-line (ZPL) at 2.99 eV is accompanied by 3 phonon repetitions
- PL temperature dependence indicates only one transition center
- DFT calculations indicate that this line originate from oxygen vacancy defect (derived from trigonal oxygen)
- PL temperature dependence and short Decay time (about 4 ns at RT) indicate that this state might be promising as a SPE center